

# ALD and CVD of Copper-Based Metallization for Microelectronic Fabrication

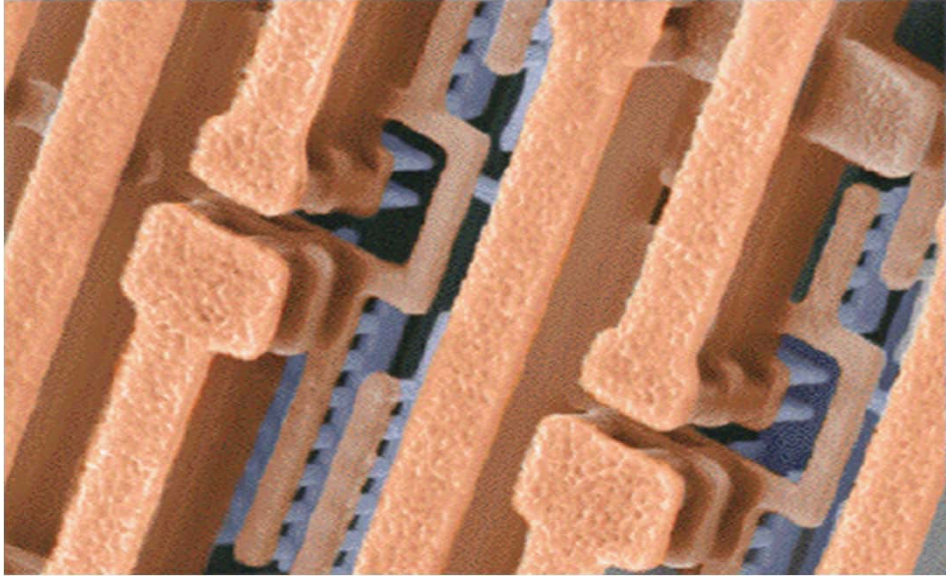
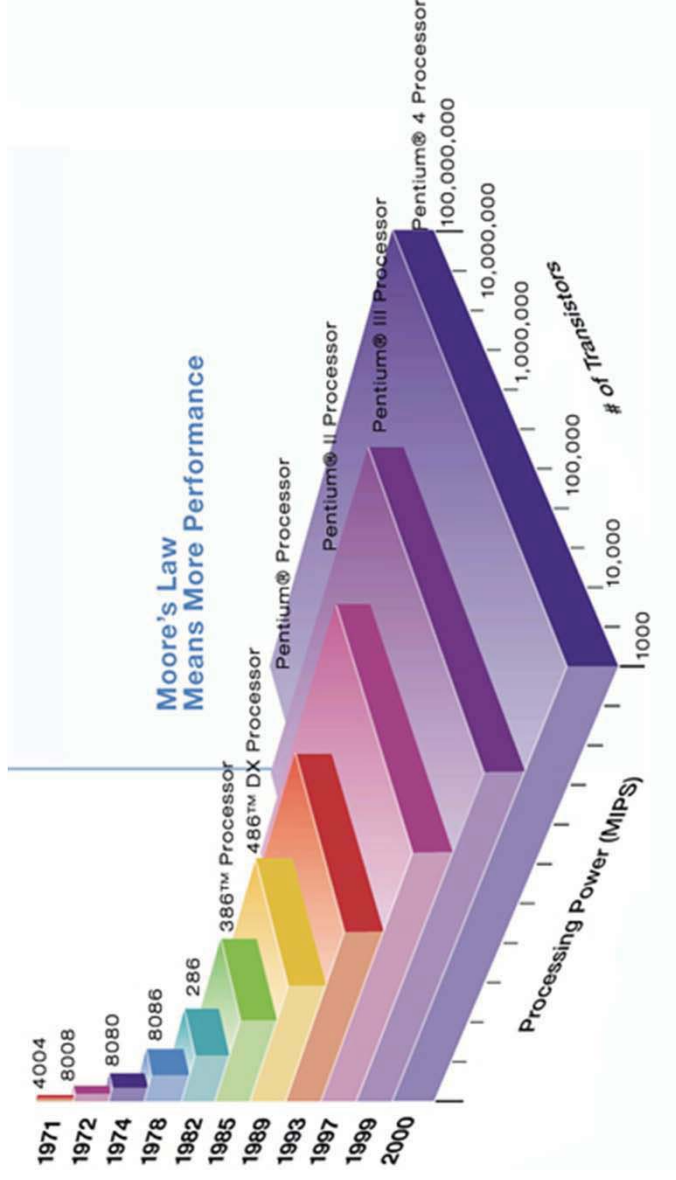


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# Introduction

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- Periodic improvements in performance of microelectronic devices have been achieved through device-scaling

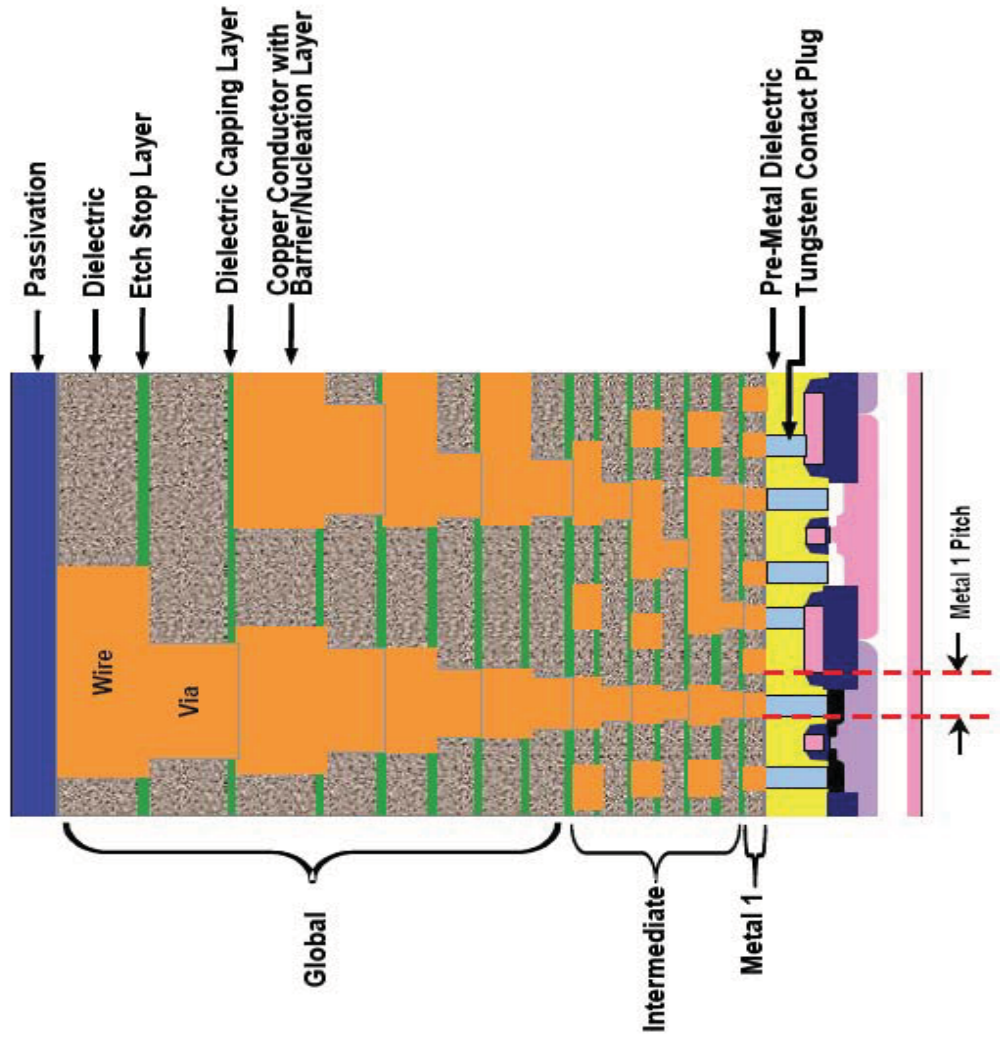


- Copper was selected because of its (1) abundance, (2) low resistivity, and (3) better electromigration reliability
- Damascene process (EP and CMP) is commonly adopted for patterning copper

# Outline

In today's presentation:

- ALD and CVD of Cu films from a Cu(I) amidinate precursor
- Formation of Cu seed layer by ALD of Cu and by CVD of CuON
- Bottom-up filling of CVD-Cu and CuMn alloy in nanoscale features
- Summary

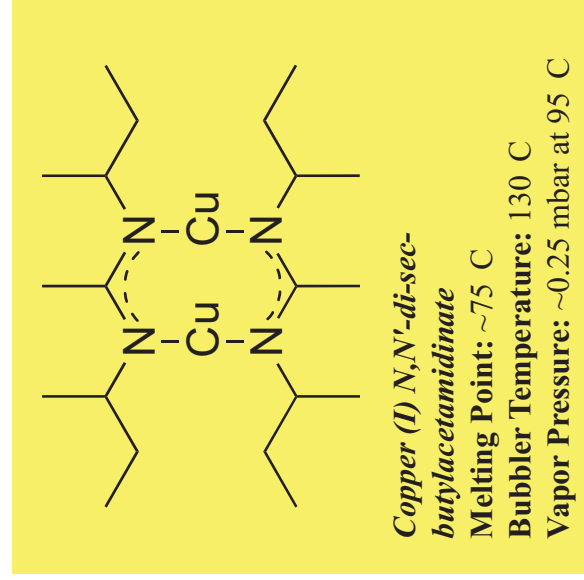
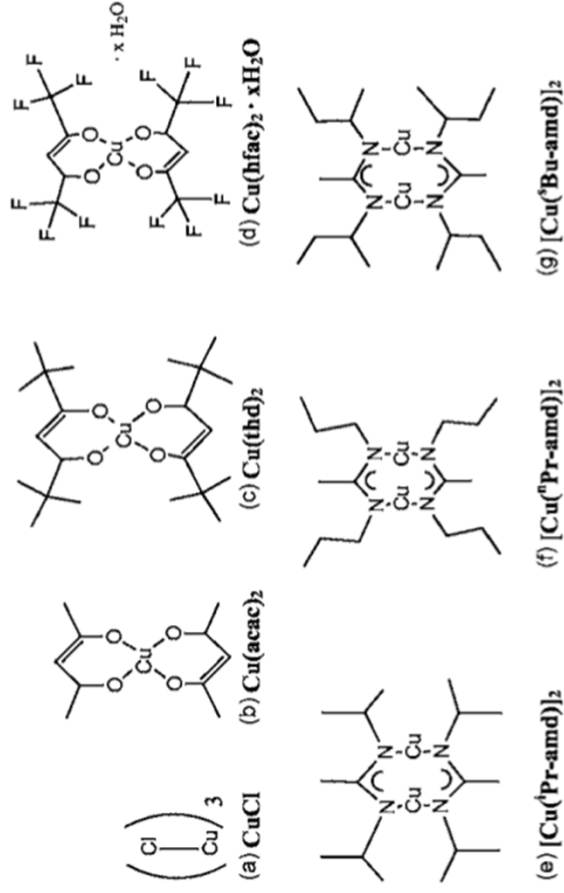


Cross-Section of Microprocessors

# Copper Precursors

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- Requirements for good ALD Cu precursors: (1) thermally stable, (2) volatile, and (3) minimal contaminations



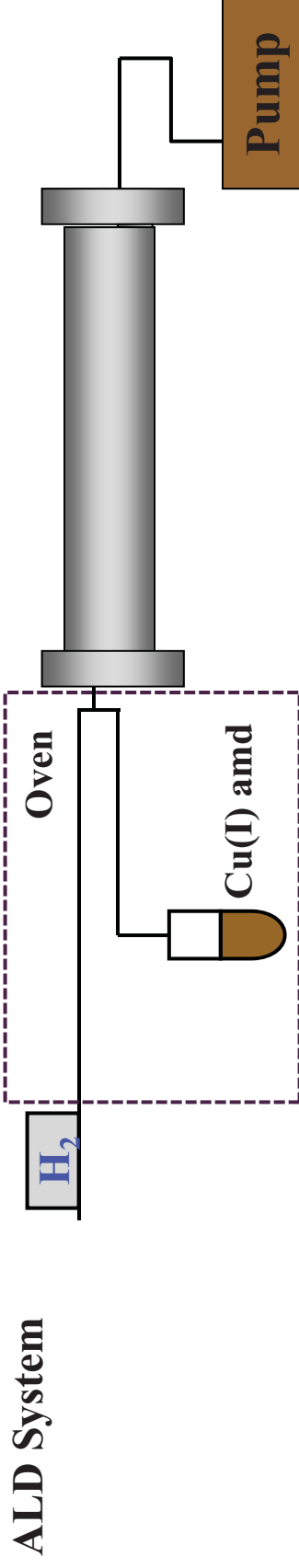
## Advantages of metal amidinates precursors:

- Bidentate chelating effect enhances thermal stability
- Tunable reactivity and volatility
- Minimal carbon and oxygen contamination

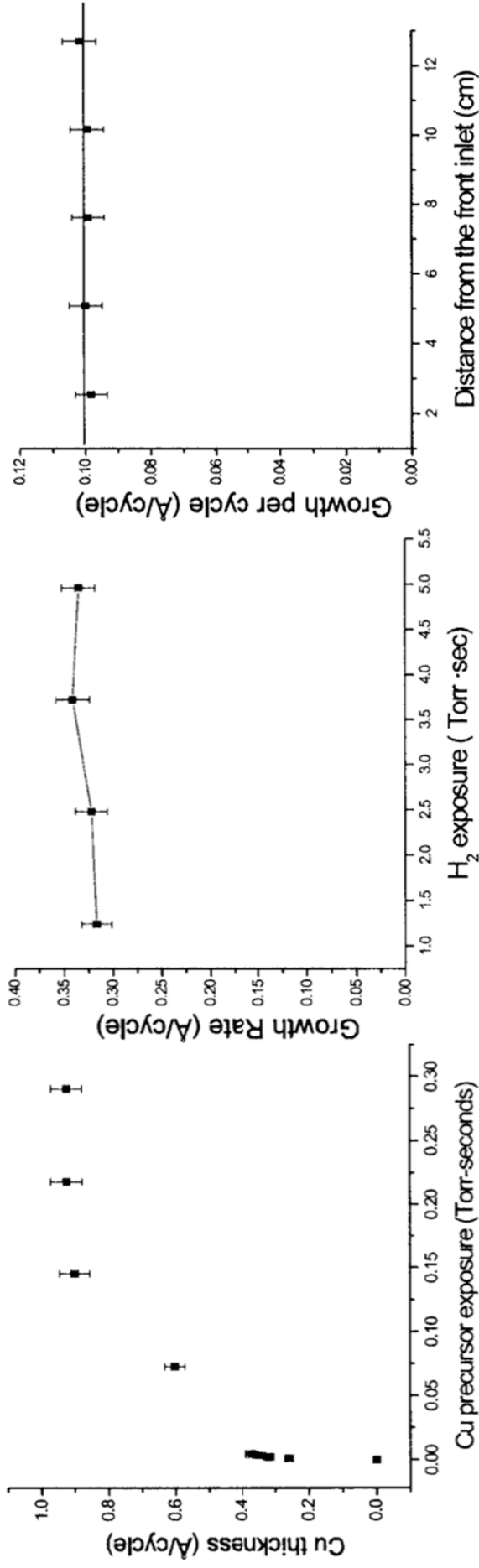
B. S. Lim, A. Rahtu, J. S. Park, and R. G. Gordon, *Inorg. Chem.*, **42** (24), 7951-7958, (2003).

# ALD of Copper

- Copper films could be deposited by ALD using molecular hydrogen as reducing agent



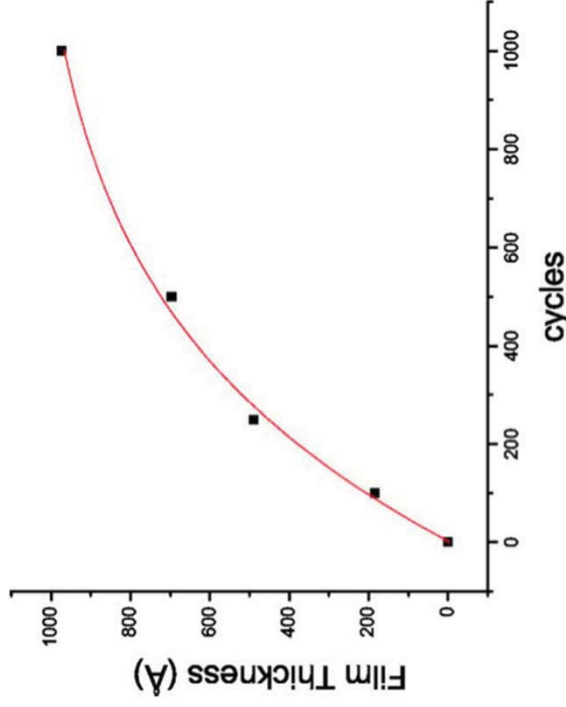
- Copper deposited on ALD- $Al_2O_3$  substrate at low temperatures (150-190 C):



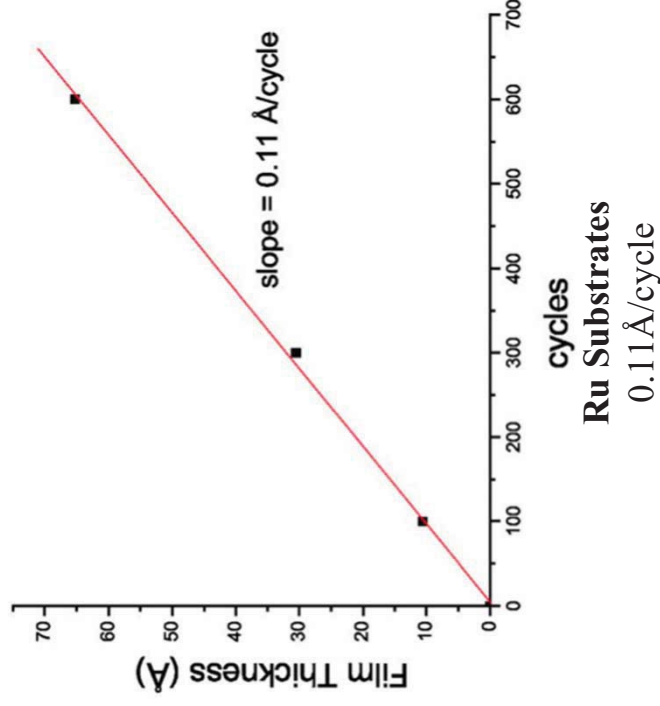
# ALD of Copper

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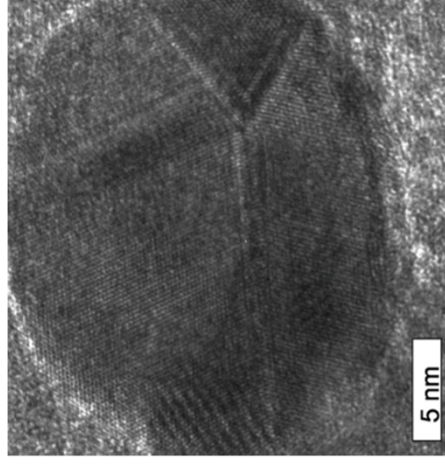
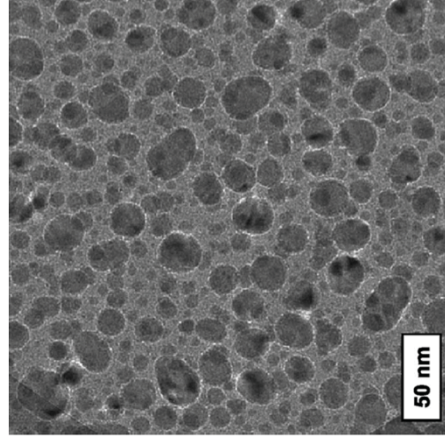
- Growth behavior can be affected by many factors: surface chemistry, precursor exposure, deposition temperature, etc.



**ALD-Al<sub>2</sub>O<sub>3</sub>, ALD-HfO<sub>2</sub>, Thermal SiO<sub>2</sub>**  
Initially ~2Å/cycle, ~0.5Å/cycle when surface is fully covered by Cu



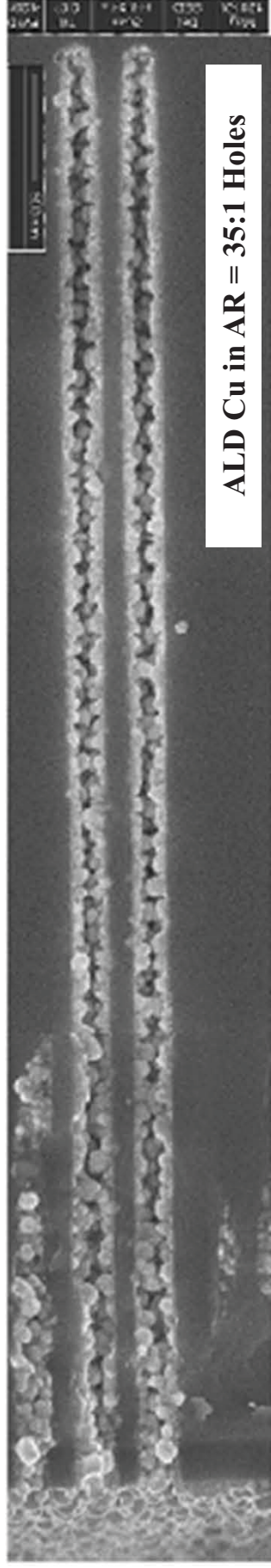
**Ru Substrates**  
0.11Å/cycle



Substrate	Growth per cycle (Å/cycle)
Al <sub>2</sub> O <sub>3</sub> /SiO <sub>2</sub>	1.90 (based on 100 cycles)
Si <sub>3</sub> N <sub>4</sub>	1.50 (based on 60 cycles)
WN	0.54 (based on 30 cycles)
Ru	0.11 (based on 100 cycles)
Co	0.40 (based on 30 cycles)
Cu	~0.5 (from Al <sub>2</sub> O <sub>3</sub> curve)

# Copper Seed Layer Using ALD

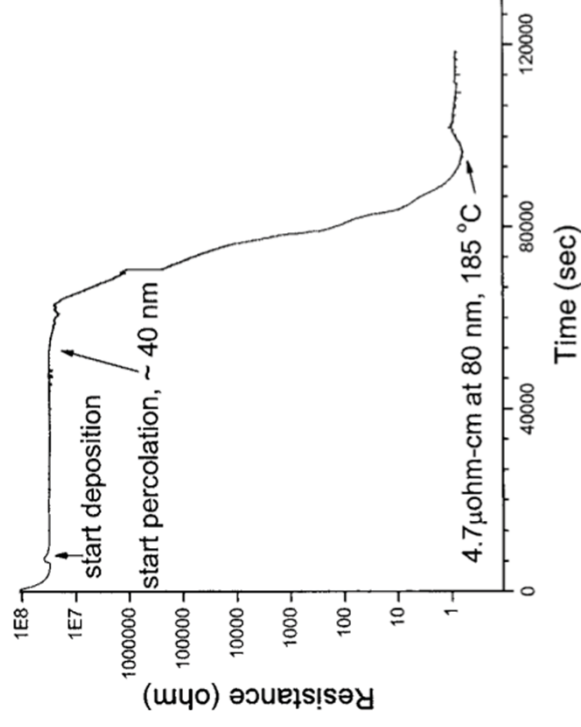
- ALD has the ability to grow films conformally and uniformly over high aspect ratio holes and trenches



- Four-point bend experiment showed high adhesion energies for Cu/Co/WN/SiO<sub>2</sub>

Structure	Scotch tape test	Adhesion energy (J/m <sup>2</sup> )
Co/SiO <sub>2</sub>	Failed	2 <sup>a</sup>
Cu/SiO <sub>2</sub>	Failed	
Cu/WN/SiO <sub>2</sub>	Failed	6 <sup>a</sup>
TaN/SiO <sub>2</sub>	Passed	>31
WN/SiO <sub>2</sub>	Passed	>31
Co/WN/SiO <sub>2</sub>	Passed	>31
Cu/Co/WN/SiO <sub>2</sub>	Passed	>31

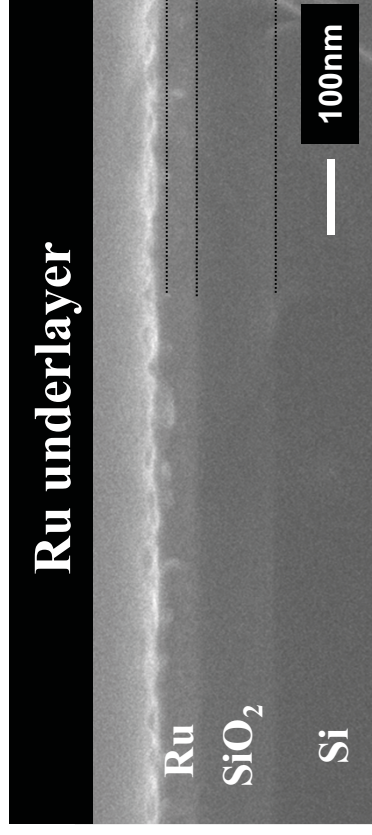
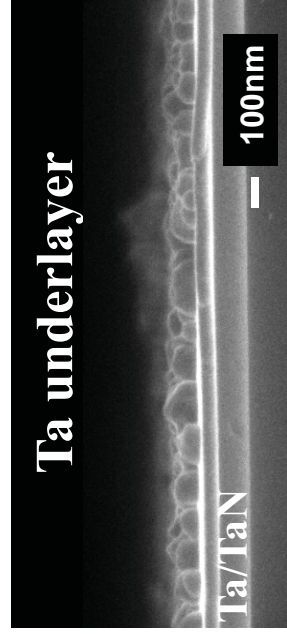
## In-situ Resistance Measurement ALD Cu on Glass (185 °C)



# Cu Seed Layer Using CVD-CuON and Plasma Reduction

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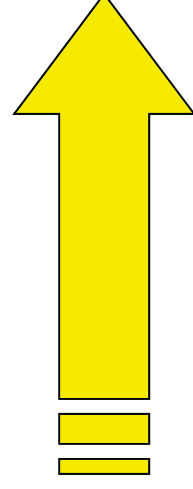
- Copper seed layers must have conformal step coverage, strong adhesion and smooth surface morphology



- Island growth of CVD-Cu on Ta underlayer
- Cu has fairly high wettability on Ru, but requires >20nm to form a continuous film due to island growth
- New approach:



**Low Surface Energy** (22-26 mJ/m<sup>2</sup> for Cu<sub>2</sub>O and Cu<sub>3</sub>N, compared to 1700-1900 mJ/m<sup>2</sup> for Cu)



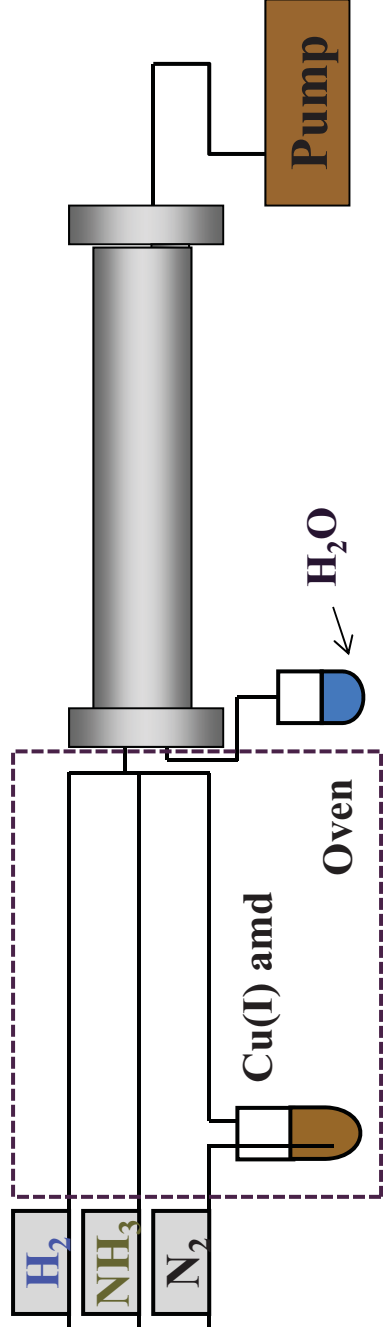
Remote Hydrogen Plasma  
Reduction near RT

Thin (<10 nm), Smooth  
(RMS ~1 nm),  
High Density (95%) CVD  
Cu Seed Layer

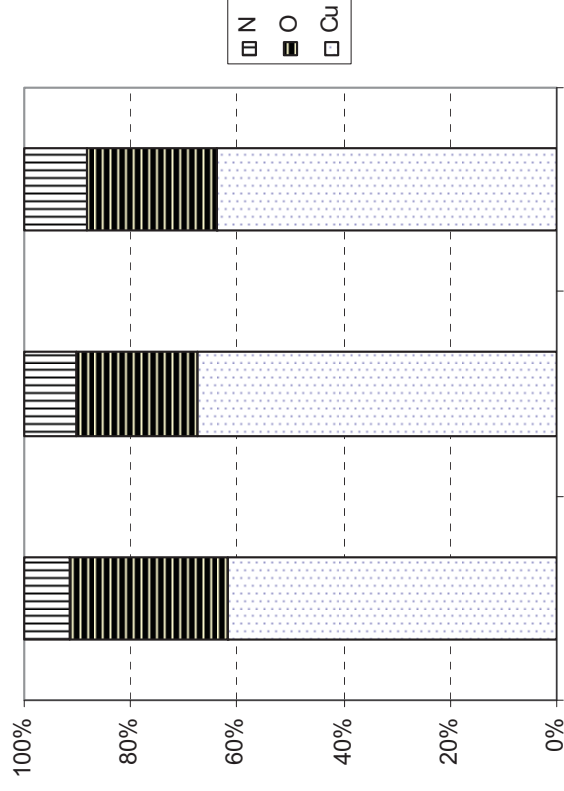
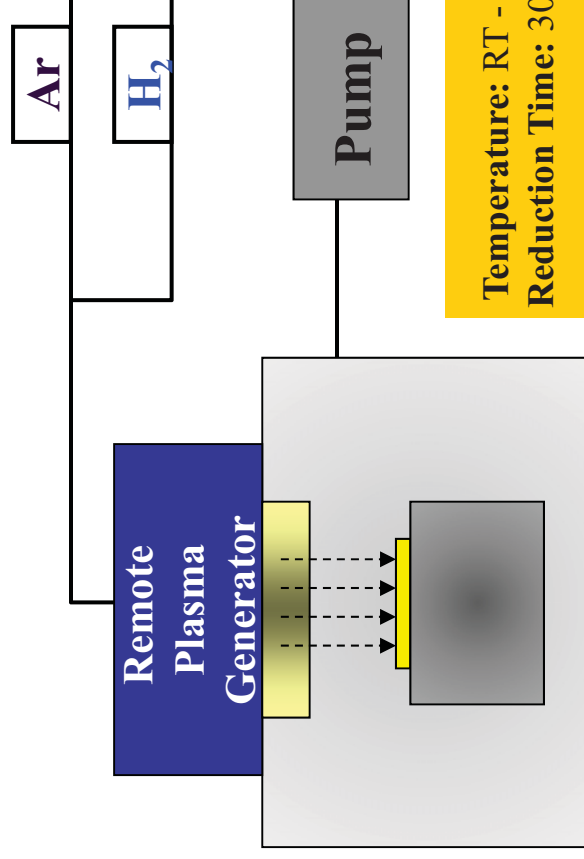
# Cu Seed Layer Using CVD-CuON and Plasma Reduction

## CVD System

Temperature: 140-220°C  
Pressure: 8 Torr



## Plasma System



Composition of CVD-CuON Films

(H<sub>2</sub>O:NH<sub>3</sub>=30:10)

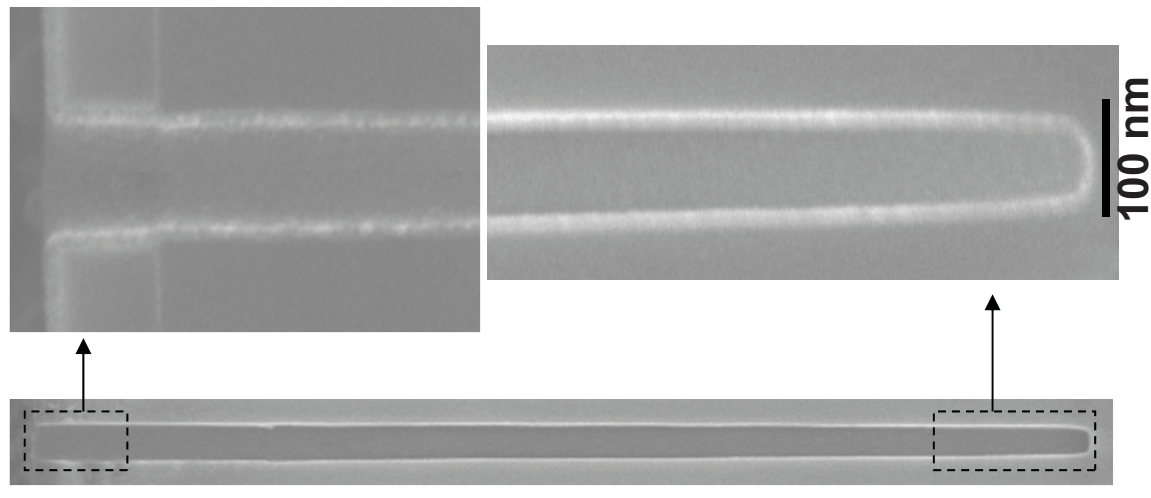
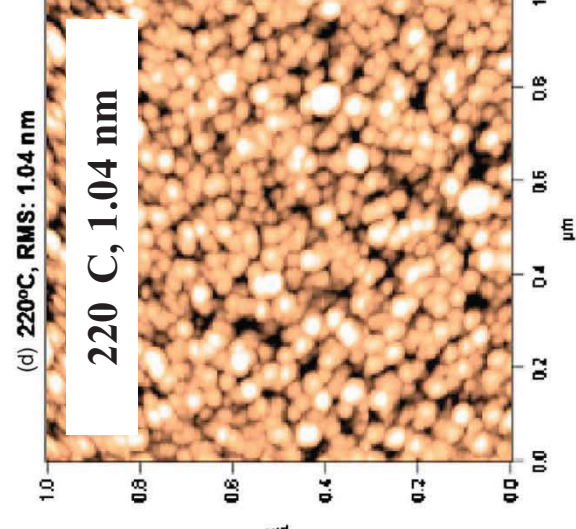
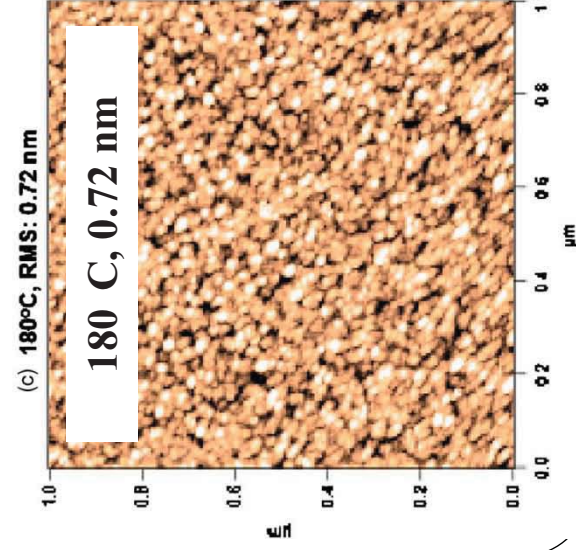
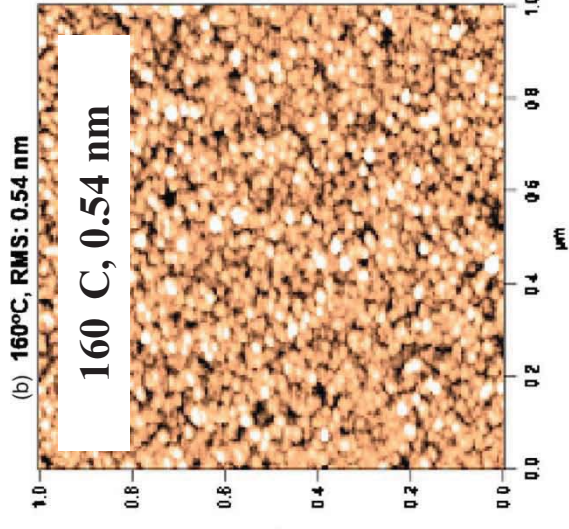
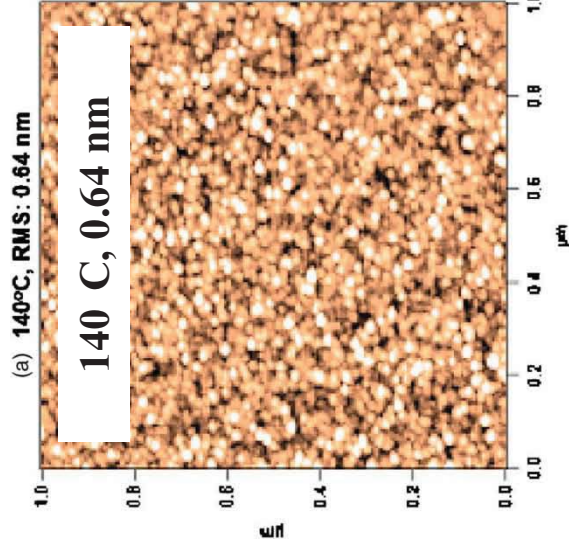
# Cu Seed Layer Using CVD-CuON and Plasma Reduction

## Surface Morphology of 20nm of CVD-CuON Films

## Step Coverage in High AR Holes

( $H_2O:NH_3=30:10$ )

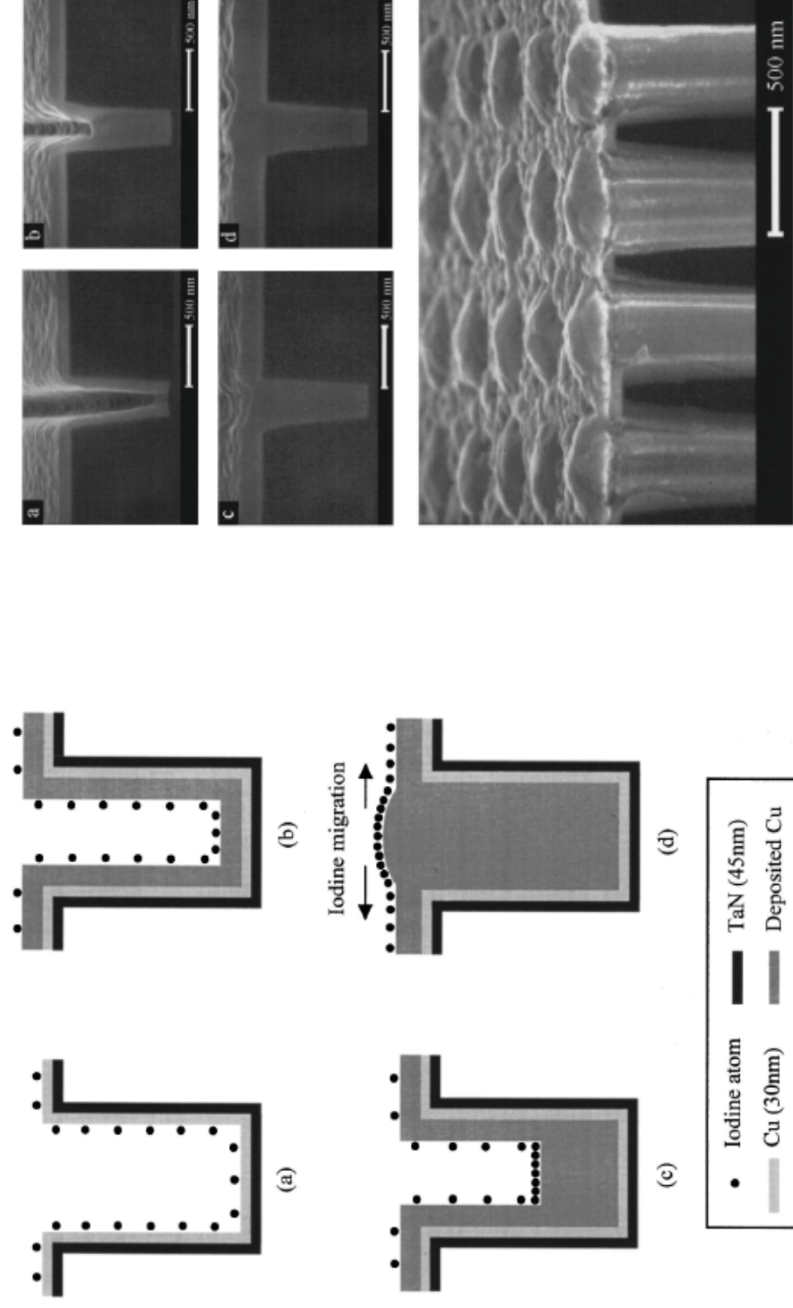
( $H_2O:NH_3=30:10, 140\text{ C}$ )



## Filling Narrow Features with CVD of Copper

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- Conventional techniques lead to formation of voids and seams in very narrow features
- Iodine is a catalytic surfactant that promotes smoother morphology and higher deposit rate
- Bottom-up filling of sub-micrometer features could be achieved by CVD

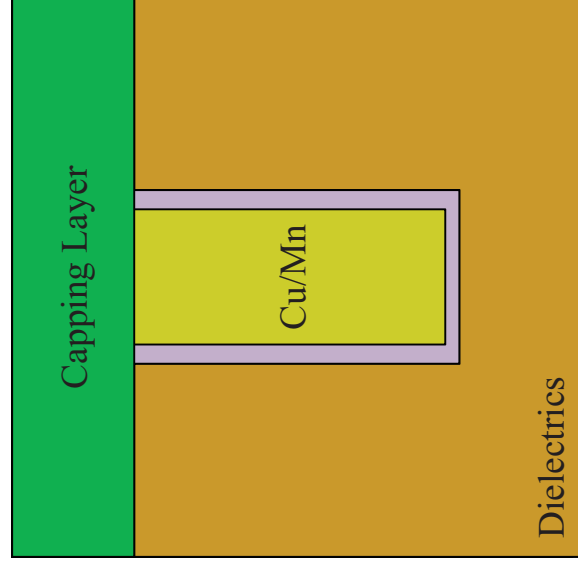


- This process requires a conformal Cu seed layer on top of the diffusion barrier and adhesion layer

E. S. Hwang and J. Lee, *Chem. Mater.*, **12**, 2076 (2000).  
K. Shim, O. Kwon, H. Park, W. Koh, and S. Kang, *J. Electrochem. Soc.*, **149** (2) G109-G113 (2002).

# Surfactant Catalyzed CVD Cu and CuMn in Narrow Trenches 12

## Motivation



## Key Points

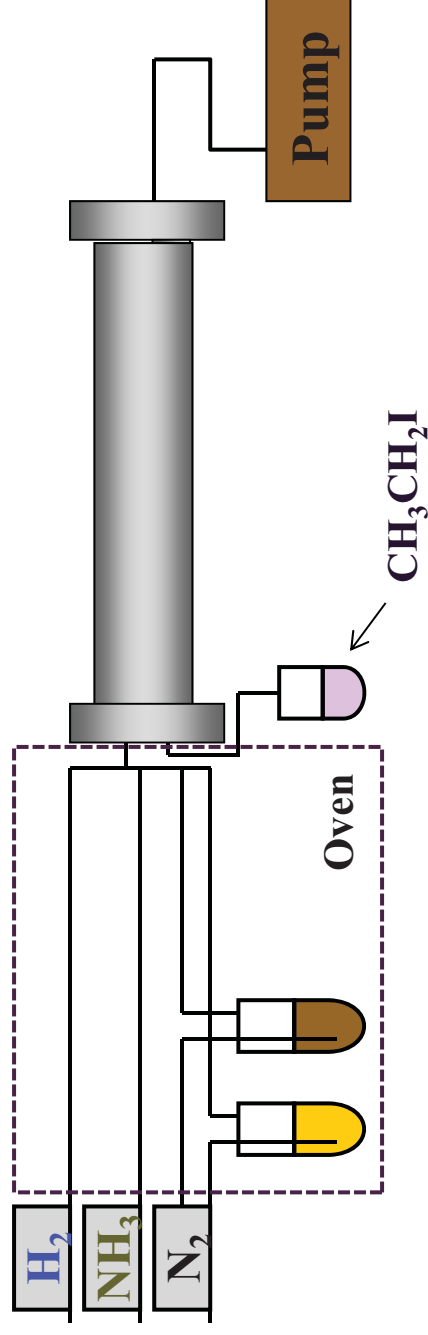
- Conformally deposited manganese nitride serves as a barrier/adhesion layer
- Iodine acts as a surfactant catalyst to promote Cu and Mn growth
- Void-free, bottom-up filling of Cu or Cu-Mn alloy in narrow trenches with AR up to at least 5:1
- Mn diffuses out from Cu during post-annealing to further improve adhesion and barrier properties at Cu/insulator interface

# Chemical Vapor Deposition of Copper

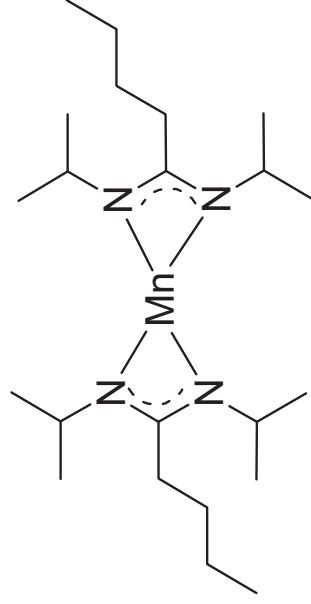
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## CVD System

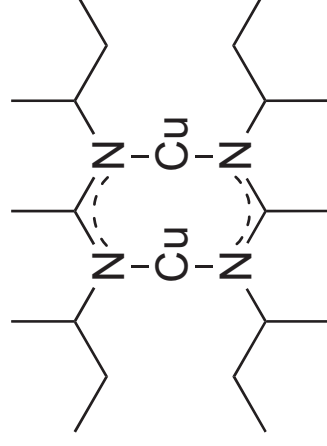
**Temperature**  
130°C for  $\text{Mn}_4\text{N}$   
180°C for Cu and CuMn  
**Pressure: 5 Torr**



## Precursors



**Bis(*N,N'*-diisopropylpentylamido)manganese(II)**  
Melting Point:  $\sim 60^\circ\text{C}$   
Bubbling Temperature:  $90^\circ\text{C}$   
Vapor Pressure:  $\sim 0.1$  mbar at  $90^\circ\text{C}$

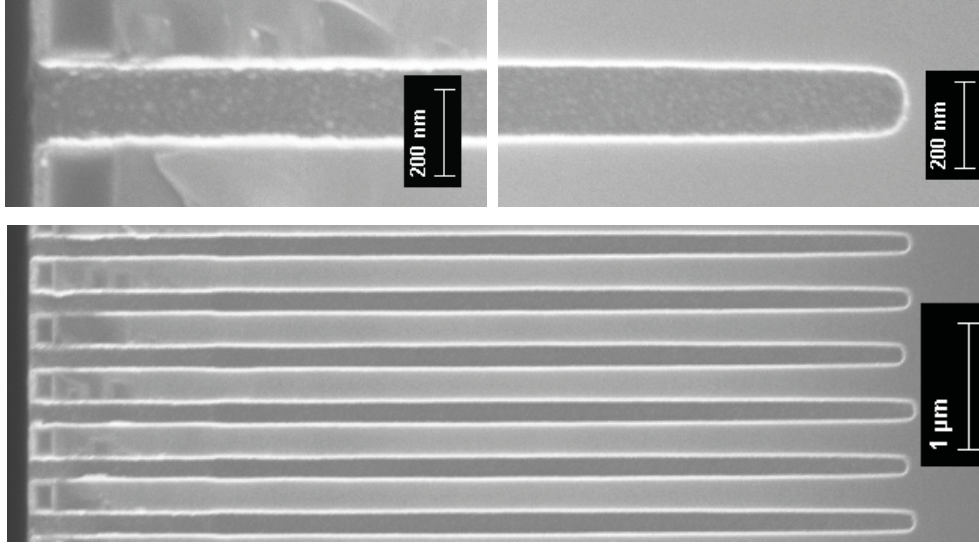
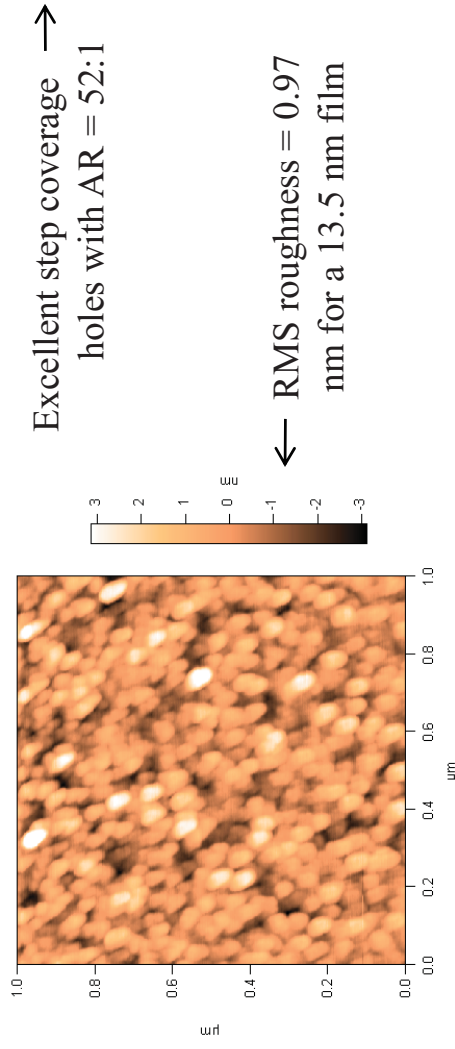
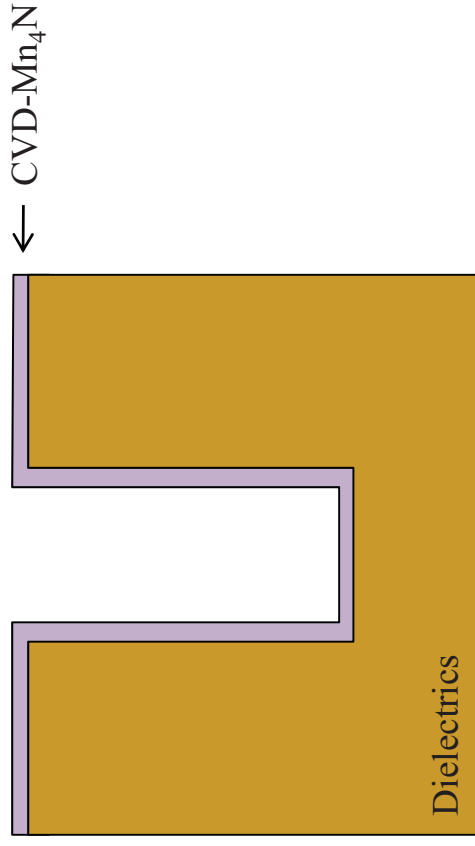


**Copper(I) *N,N'*-di-sec-butylacetamidate**  
Melting Point:  $\sim 75^\circ\text{C}$   
Bubbling Temperature:  $130^\circ\text{C}$   
Vapor Pressure:  $\sim 0.25$  mbar at  $95^\circ\text{C}$

## CVD-Mn<sub>4</sub>N Barrier/Adhesion Layer

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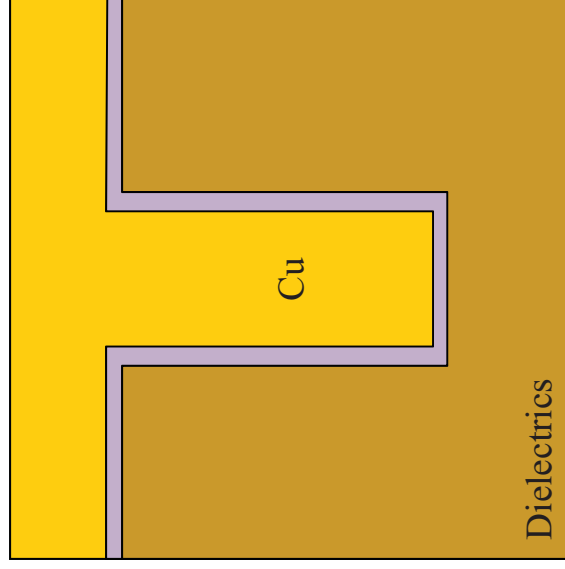
- CVD-Mn<sub>4</sub>N ( $\epsilon$  phase, FCC structure) can be prepared by reacting manganese amidinate precursors with NH<sub>3</sub>



- Mn<sub>4</sub>N layer as thin as 2.5 nm (1) shows barrier properties against Cu diffusion, (2) significantly improve adhesion (debonding energy = 6.5 J/m<sup>2</sup>) between Cu and SiO<sub>2</sub>
- Release of iodine and catalytic effects are observed on Mn<sub>4</sub>N underlayer

# Surfactant Catalyzed Bottom-up Filling of CVD-Cu

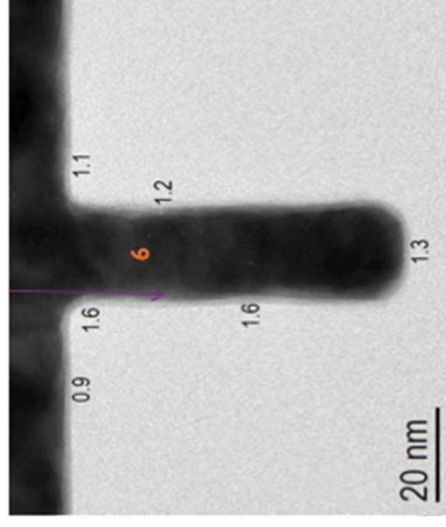
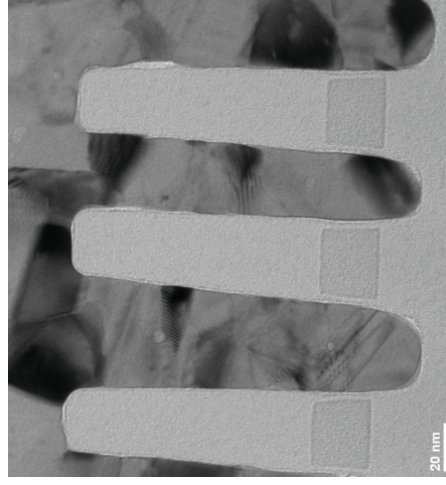
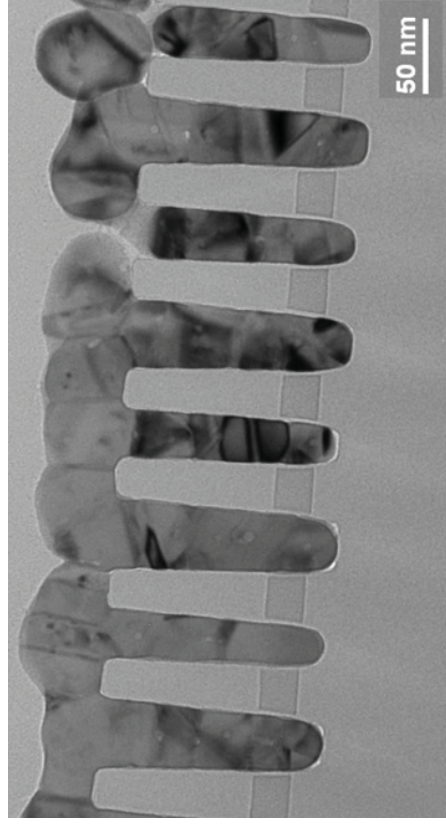
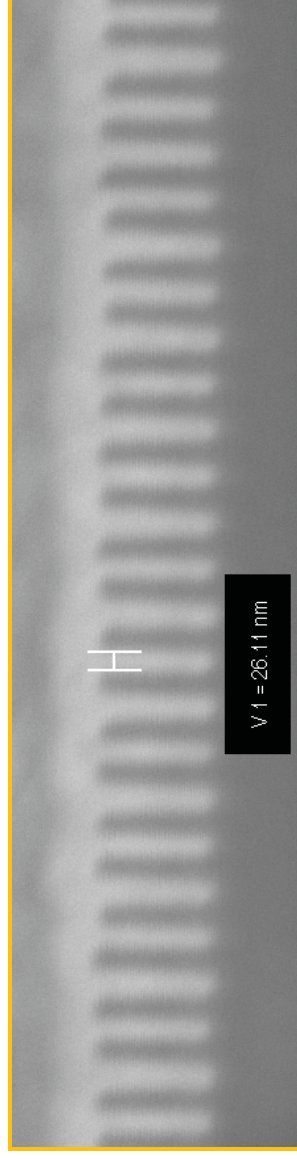
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CVD-Mn<sub>4</sub>N  
Deposition

Iodine  
Exposure

CVD-Cu  
Deposition



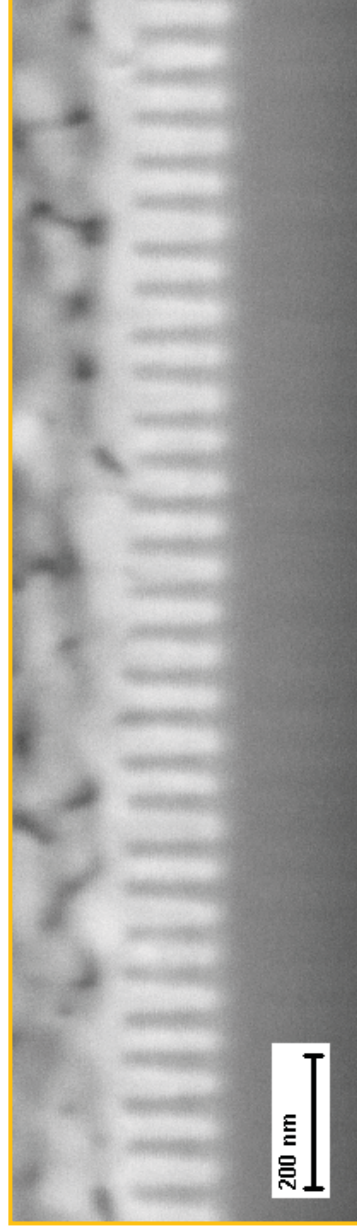
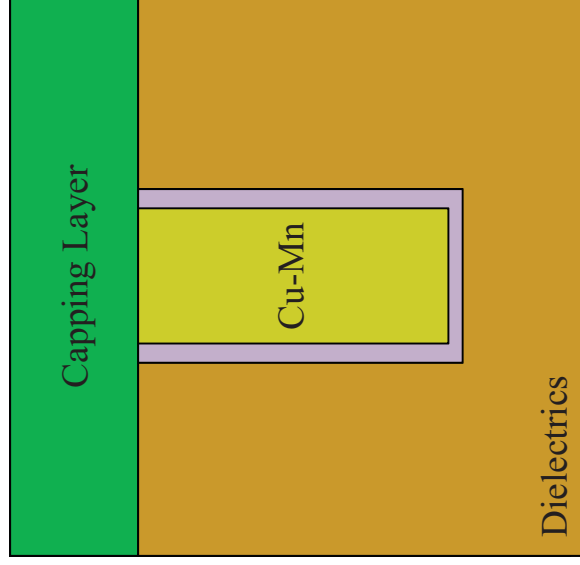
26 nm Structure

18 nm Structure

**With CVD-Mn<sub>4</sub>N liner layer and iodine catalyst, trenches with width  $\leq 20$  nm and aspect ratio over 5:1 can be completely filled with CVD-Cu**

## Surfactant Catalyzed Bottom-up Filling of CVD-CuMn Alloy 16

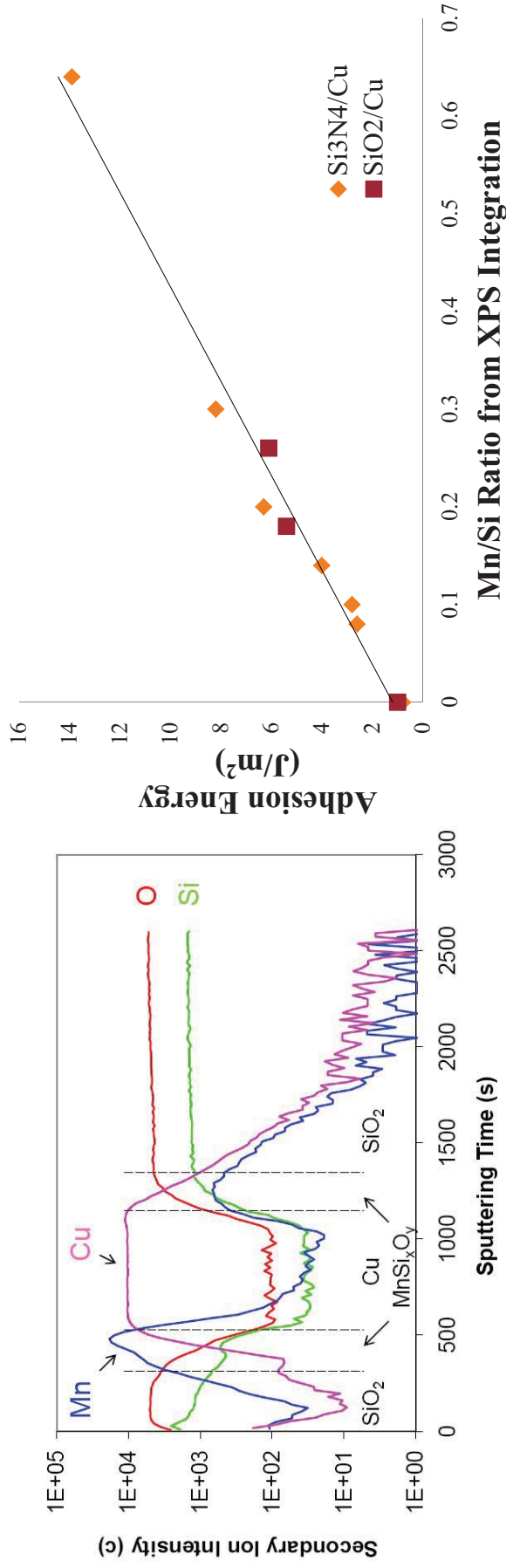
- Cu-Mn alloy can be formed by (1) alternating CVD-Cu and Mn or (2) co-depositing Cu and Mn



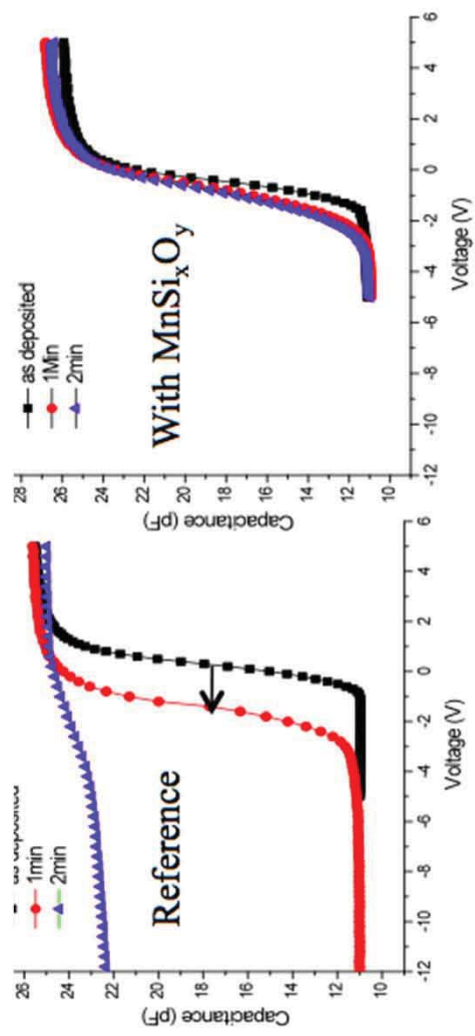
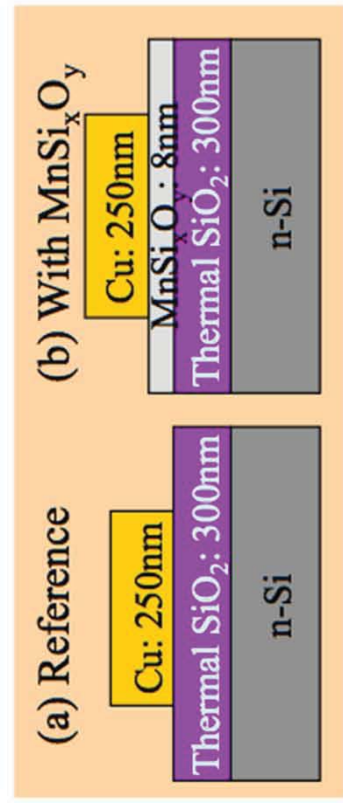
**Trenches with width  $\leq 30$  nm can be completely filled with CuMn alloy**  
Manganese concentration: 0.5-2.0 atomic %

# Enhancement by Diffusion of Mn from Cu to Interface

- Insulators encourages diffusion of Mn through Cu grain boundaries to interface
- Mn improves both adhesion and barrier properties at the interface



## Cu Diffusion Barrier Test



## Summary

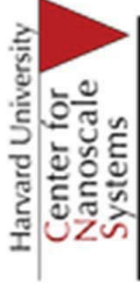
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- ✓ Copper can be deposited by ALD or CVD using a Cu(I) amidinate precursor
- ✓ Conformal and uniform seed layers can be prepared by ALD-Cu or by CVD-CuON followed by remote hydrogen plasma reduction
- ✓ Nanoscale trenches can be superconformally filled by CVD-Cu and CVD-CuMn alloy with an iodine surfactant on Mn<sub>4</sub>N liner layer
- ✓ Manganese in Cu-Mn alloy diffuses out to strengthen the interface between Cu and insulators without increasing the resistivity of Cu
- ✓ Manganese silicate (MnSi<sub>x</sub>O<sub>y</sub>) interfacial layer shows excellent barrier properties against Cu diffusion and protects Cu from corrosion by H<sub>2</sub>O and O<sub>2</sub>

## Acknowledgements

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- Facilities at Harvard's Center for Nanoscale Systems (CNS), a member of the National Nanotechnology Infrastructure Network (NNIN), supported by the National Science Foundation



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Substrates and Analyses: Applied Materials, IMEC and IBM



- Members of Gordon Group

